SYNTHESIS AND CHARACTERIZATION OF THE SURFACE MODIFIED TITANIUM DIOXIDE/EPOXY NANOCOMPOSITES

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Epoxy resins, one of the most important thermosetting polymers, have been extensively used in coatings, electronic materials, adhesives, as matrices for advanced composites, and so on [1-3]. Among different inorganic fillers used to improve the properties of polymer matrices, titanium dioxide is one of the most investigated [4-5]. However, commercial micrometer size inorganic pigments could have the adverse effect on the properties of crosslinked coatings, especially on adhesion of the coating to the substrate, can reduce the elasticity of the coating, its impact resistance, transparency, and resistance to abrasion and scratching and can cause delamination of the coating [6-9].

Because of the small size, nanoparticles have a high specific surface area and their presence can improve mechanical, optical, thermal and barrier properties of the epoxy coatings [10-14]. Such property improvement can be achieved by uniform distribution of nanoparticles through the polymer matrix, i.e. by preventing the aggregation of nanoparticles which can occur due to their high interfacial reactivity and high surface tension energy, and by improving the compatibility between nanoparticles and polymer, which is usually accomplished after the adequate surface modification of nanoparticles. The most common way to modify the surface of nanoparticles, control nanoparticles aggregation and stabilize the colloidal nanoparticles is to attach suitable organic groups to the surface atoms of nanoparticles [15-17]. After the appropriate surface modification, nanoparticles become hydrophobic, which can decrease their surface tension and lead to better miscibility and compatibility of nanoparticles with the polymer matrix [18]. Organic groups that can be used to attach ligands to the titanium dioxide surface are amine [19], ammonium [20] and different benzene derivatives, mostly catechol and salicylic acid [21-23]. Aromatic ligands with two or three adjacent phenolic OH groups (catechol, pyrogallol and gallic acid) can be adsorbed on the surface of TiO2 by forming chelating or bridging complexes with titanium ions on the surface [24-25]. In our previous works, TiO2 nanoparticles were surface modified with different alkyl gallates [26] and then incorporated into the poly(methyl methacrylate) and epoxy matrix [27]. The length of the aliphatic part of the used gallates showed a great effect on dispersability of the surface modified TiO2 nanoparticles and on the properties of both matrices.

In this article, surface modification of TiO2 nanoparticles was performed with three gallic acid esters with different lengths of alkyl chains (octyl (8C), decyl (10C) and lauryl (12C) gallate). Surface modified TiO2 nanoparticles were characterized using Fourier transform infrared (FTIR) and ultraviolet–visible (UV–Vis) spectroscopy. The influence of different gallic acid esters used for surface modification and content of modified TiO2...
nanoparticles on the glass transition temperature, rheological, barrier and mechanical properties of epoxy resin was investigated.

**Materials and methods**

**Materials**
Titanium isopropoxide was obtained from TCI Europe. Gallic acid, octyl gallate (OG), lauryl gallate (LG), decyl alcohol and 2-propanol were obtained from Sigma–Aldrich. Decyl gallate (DG) was synthesized by esterification of gallic acid with decyl alcohol using the same synthetic procedure described elsewhere [28]. Epoxy resin (ER), CHS-EPOXY 210 X 75, was purchased from Spolchemie. The curing agent, EPIKURE 3115 X 70 was purchased from Momentive. All chemicals were used as received without further purification.

Preparation and surface modification of TiO$_2$ nanoparticles

The TiO$_2$ colloid was prepared by hydrolysis of titanium isopropoxide at 80 °C under a stream of dry nitrogen for 8 h, according to the procedure described in literature [27, 29]. The surface modification of TiO$_2$ colloidal nanoparticles with octyl gallate (TiO$_2$-OG), decyl gallate (TiO$_2$-DG), and lauryl gallate (TiO$_2$-LG) was performed in the same manner. As an example, the modification of TiO$_2$ nanoparticles with LG is described. Lauryl gallate (0.1136 g) was dissolved in 15 ml of the chloroform and methanol mixture (4:1 vol./vol.). TiO$_2$ colloid solution (2 ml) was diluted ten times with deionized water and mixed with the prepared lauryl gallate solution in a separator funnel. After shaking for a short time, a red chloroform phase, containing TiO$_2$ particles surface modified with lauryl gallate (TiO$_2$-LG), separated from the aqueous phase. The obtained red chloroform solution was drop-wise added to a 100 times larger volume of methanol to remove residual free lauryl gallate molecules. The surface modified TiO$_2$ particles with lauryl gallate were separated by centrifugation and redispersed in chloroform for further usage.

Preparation of TiO$_2$/epoxy nanocomposite films

The commercially available epoxy resin CHS-EPOXY 210 X 75 was used as polymer matrix and cured with EPIKURE 3115 X 70 hardener using a weight ratio of 100:35. The nanocomposites with 1 wt.% of TiO$_2$, (TiO$_2$-OG1.0/ER, TiO$_2$-DG1.0/ER and TiO$_2$-LG1.0/ER), by the total weight of epoxy resin and its hardener, were prepared by adding the appropriate amount of the surface modified TiO$_2$ nanoparticles (TiO$_2$-OG, TiO$_2$-DG and TiO$_2$-LG, respectively) dispersed in chloroform to epoxy resin and after that a corresponding amount of curing agent EPIKURE 3115 X 70 was added. The obtained mixture was mixed in the ultrasonic bath (Sonorex Digitec) for 10 min. The nanocomposites with 0.5 and 2 wt.% of TiO$_2$-LG (TiO$_2$-LG0.5/ER and TiO$_2$-LG2.0/ER) were prepared in the same manner by adding the appropriate amount of TiO$_2$-LG nanoparticles to the epoxy resin. In order to prepare films of uniform thickness, the dispersions were drawn in films on glass plates with wire-wound rods. The dispersions were also cured in Teflon molds (dimensions $63 \times 12 \times 2$ mm) for DMA measurements. All samples were cured at room temperature for 21 days.

Characterization of nanoparticles

The size and shape of TiO$_2$ nanoparticles were determined by the transmission electron microscopy (TEM, JEOL 1200EX). The FTIR spectra of dry, unmodified and modified TiO$_2$ nanoparticles were recorded on a FTIR spectrophotometer (Bomem MB 100) in the form of KBr pellets. The absorption spectra of unmodified and modified TiO$_2$ and prepared nanocomposites were recorded on a UV-Vis spectrophotometer (Perkin-Elmer Lambda-5).

Characterization of nanocomposites

Differential scanning calorimetry (DSC) was used to determine the glass transition temperature, $T_g$, of epoxy film and polymer nanocomposites. The samples for DSC measurements were cut from the films cured on the glass plates. The films were pulled off after staying overnight in water and then dried in the vacuum oven for 2 h at 100 °C. All measurements were performed on a Perkin Elmer DSC-2 instrument in a nitrogen atmosphere, at the heating rate of 20 °C/min.

The water vapor transfer through the polymer films was determined by the standard gravimetric method (ASTM D1653-13) using the permeability test dish (BYK-Gardner permeability cup). The desiccant method was used and measurements were carried out under the conditions of constant relative humidity of 85%. The water vapor transmission rate (WVTR) was calculated using the equation 1:

$$WVTR = \frac{\Delta m}{tA}$$

where $A$ is the surface area of the polymer film through which the water vapor can pass during the time $t$, while $\Delta m$ is the mass of transported water.

The dependence of the storage modulus, loss modulus and the loss factor on temperature at a constant frequency was determined by the dynamic mechanical analysis (DMA) using TA Instruments SDT Q600 operating in the temperature sweep mode. All measurements were performed in the temperature range 25 -150 °C, at a frequency of 1 Hz, and deformation of 0.1%.

The thickness of TiO$_2$-epoxy nanocomposite dry film was determined by thickness gauge (Byko-test 4500 Fe/ NFe, BYK-Gardner). Hardness of nanocomposites was determined by the pendulum hardness tester using König pendulum (BYK-Gardner). Hardness testing was performed in three different places on the coating and the average value of the number of oscillations was used as the hardness of the film.
Results and discussions

Characterization of the TiO$_2$ nanoparticles

TiO$_2$ nanoparticles in the anatase crystalline form were prepared by acid catalyzed hydrolysis of titanium isopropoxide [29]. The average particle size was estimated by TEM (Figure 1), and it was found that the particles are roughly spherical in shape with the average particle size of 3.9 ± 0.9 nm [27].

![Figure 1. TEM image of TiO$_2$ nanoparticles](image1)

The surface modification of TiO$_2$ nanoparticles was performed in order to enable the nanoparticles transfer from the aqueous to the organic phase. Three gallic acid esters with different lengths of the hydrophobic part (octyl, decyl and lauryl gallate) were used for surface modification of the synthesized TiO$_2$ nanoparticles. Charge transfer complex (CTC) between the TiO$_2$ nanoparticles and formed alkyl gallates was confirmed by FTIR and UV–Vis spectroscopy. FTIR spectra of octyl gallate, dried TiO$_2$, and TiO$_2$ nanoparticles modified with octyl gallate are shown in Figure 2.

The spectrum of dried TiO$_2$ nanoparticles contains a wide band in a range from 3700 to 3000 cm$^{-1}$, assigned to the stretching vibrations of surface OH groups and symmetric and asymmetric vibrations of the surface adsorbed water. The bands between 2960 and 2850 cm$^{-1}$ are attributed to the organic residues formed during the synthesis of colloids. The band at 1630 cm$^{-1}$ corresponds to the bending vibration of adsorbed water.

In the FTIR spectrum of OG, characteristic bands at 3450 and 3350 cm$^{-1}$ correspond to the OH stretching vibrations. The bands at 2950 and 2870 cm$^{-1}$ originate from asymmetric and symmetric stretching vibrations of C─H bond from a methyl group, while the bands at 2930 and 2850 cm$^{-1}$ correspond to the asymmetric and symmetric stretching vibration of C─H bond of a methylene group. The band at 1670 cm$^{-1}$ is ascribed to stretching vibrations of ester carbonyl groups, and the bands at 1608, 1533, and 1408 cm$^{-1}$ are assigned to aromatic ring stretching vibrations. The band at 1465 cm$^{-1}$ is due to the aliphatic C─H bending and aromatic ring stretching vibrations and the band at 1379 cm$^{-1}$ is due to C─O stretching vibrations of the phenolic group. The band at 1330 cm$^{-1}$ corresponding to the in-plane OH bending vibration is overlapped with the band at 1305 cm$^{-1}$, assigned to C─O stretching vibrations of the phenolic group. Other characteristic bands at 1260, 1196 and 1030 cm$^{-1}$ can be assigned to C(═O)─O stretching vibrations of ester groups, in-plane bending vibrations of the phenolic group and O─C─C stretching vibrations of the ester group, respectively.

In the FTIR spectrum of TiO$_2$, characteristic bands with OG, characteristic bands at 3450 and 3350 cm$^{-1}$ completely disappeared. The bands corresponding to the C(═O)─O bond and O─C─C vibrations became broader and shifted to lower and higher frequencies, respectively. The band at 1379 cm$^{-1}$, corresponding to C─O stretching vibrations of the phenolic group became broader and shifted towards lower frequencies. Also, the band at 1304 cm$^{-1}$ did not disappear completely, but its intensity was drastically reduced. According to the FTIR results, one OH group of gallic acid esters remained unbounded, while OG was bonded to the surface Ti atoms through the adjacent OH groups forming a binuclear (bridging) complex. Characteristic bands observed in the FTIR spectra of TiO$_2$ nanoparticles modified with decyl and lauril gallate are similar in shape and position as in the FTIR spectra of TiO$_2$ nanoparticles modified with octyl gallate.

In addition to the results obtained from FTIR spectro-
scopy, the formation of the charge transfer complex between the surface of TiO$_2$ nanoparticles and gallic acid esters was also confirmed by UV-Vis spectroscopy. The formation of CTC between alkyl gallates and TiO$_2$ is the reason for the appearance of red color of the obtained dispersions. The absorption spectra of the aqueous TiO$_2$ solution and TiO$_2$-OG, TiO$_2$-DG and TiO$_2$-LG solutions in chloroform are given in Figure 3. The absorption spectra of TiO$_2$ nanoparticles surface modified with gallates are shifted to higher wavelengths in comparison to the absorption spectrum of aqueous TiO$_2$. The absorption of the surface modified TiO$_2$ nanoparticles starts at around 650 nm and the absorption of aqueous TiO$_2$ starts at around 380 nm. This significant shift of the absorption spectra indicates the reaction of gallic acid esters with surface Ti atoms of nanoparticles and the formation of CTC [26, 30]. However, the obtained results revealed that the hydrophobic part length of gallates used for the surface modification of TiO$_2$ nanoparticles had no significant influence on the position of the absorption spectra of TiO$_2$ nanoparticles.

![Figure 3. The absorption spectra of the aqueous TiO$_2$ colloid solution and solutions of TiO$_2$-OG, TiO$_2$-DG and TiO$_2$-LG in chloroform](image)

UV-Vis spectroscopy of TiO$_2$/epoxy nanocomposites

The presence of the surface modified TiO$_2$ nanoparticles in the epoxy matrix was investigated by UV-Vis spectroscopy. The absorption spectra of pure epoxy resin and prepared TiO$_2$/epoxy nanocomposites films are presented in Figure 4. From the obtained results it can be observed that surface modified TiO$_2$ nanoparticles are present in an unchanged form in the epoxy matrix. The increase of the hydrophobic part length of gallates used for the surface modification of TiO$_2$ nanoparticles induced red shift of the nanocomposites absorption spectra at higher wavelengths (Figure 4a). Also, the increase of the TiO$_2$-LG content shifted the absorption spectra of prepared nanocomposites to higher wavelengths (Figure 4b).

![Figure 4. The absorption spectra of epoxy resin and a) TiO$_2$-OG1.0/ER, TiO$_2$-DG1.0/ER and TiO$_2$-LG1.0/ER and b) TiO$_2$-LG0.5/ER, TiO$_2$-LG1.0/ER and TiO$_2$-LG2.0/ER nanocomposites](image)

DSC analysis of the synthesized TiO$_2$/epoxy nanocomposites

The influence of the surface modified TiO$_2$ nanoparticles on the glass transition temperature of TiO$_2$/epoxy nanocomposites was determined from the DSC results. DSC thermograms of crosslinked neat epoxy resin, nanocomposites containing 1.0 wt.% of the surface modified TiO$_2$ nanoparticles and nanocomposites prepared with different contents of lauryl gallate are shown in Figure 5. The glass transition temperature of the investigated samples was obtained as the peak maximum of the heat flow first derivative and the determined values are shown in Table 1.
Figure 5. DSC thermograms of the examined epoxy resin and a) TiO$_2$-OG1.0/ER, TiO$_2$-DG1.0/ER and TiO$_2$-LG1.0/ER and b) TiO$_2$-LG0.5/ER, TiO$_2$-LG1.0/ER and TiO$_2$-LG2.0/ER nanocomposites

Incorporation of the surface modified TiO$_2$ nanoparticles led to the increase of $T_g$ of epoxy resin, indicating the presence of attractive interactions between TiO$_2$ nanoparticles and the polymer matrix and the reduction of the mobility of segments at the interface between the polymer matrix and TiO$_2$ nanoparticles [31-33]. Also, curing reactions may cause partial aminolysis of gallate ester groups and amide formation, which led to the chemical bonding of nanoparticles to the polymer matrix and the increase of the $T_g$ value. According to the obtained results it can be concluded that the chain length of the hydrophobic part of gallates used for surface modification of TiO$_2$ nanoparticles has no influence on the $T_g$ value, while the increase of the TiO$_2$-LG content induced a slight increase of $T_g$.

Table 1. Values of the glass transition temperature determined by DSC analysis, $T_g$ (DSC), and DMA, $T_g(G''\tan \delta)$, and values of $G'$ at 120 °C of the examined samples

<table>
<thead>
<tr>
<th>Sample</th>
<th>$T_g$ (DSC), °C</th>
<th>$T_g(G'\tan \delta)$, °C</th>
<th>$G'$, MPa (120 °C)</th>
</tr>
</thead>
<tbody>
<tr>
<td>ER</td>
<td>79</td>
<td>66</td>
<td>87</td>
</tr>
<tr>
<td>TiO$_2$-OG1.0/ER</td>
<td>85</td>
<td>71</td>
<td>89</td>
</tr>
<tr>
<td>TiO$_2$-DG1.0/ER</td>
<td>84</td>
<td>69</td>
<td>89</td>
</tr>
<tr>
<td>TiO$_2$-LG0.5/ER</td>
<td>80</td>
<td>69</td>
<td>88</td>
</tr>
<tr>
<td>TiO$_2$-LG1.0/ER</td>
<td>83</td>
<td>69</td>
<td>87</td>
</tr>
<tr>
<td>TiO$_2$-LG2.0/ER</td>
<td>84</td>
<td>71</td>
<td>91</td>
</tr>
</tbody>
</table>

DMA of synthesized TiO$_2$/epoxy nanocomposites

A dynamic mechanical analysis of the prepared samples was performed to determine the influence of the surface modified TiO$_2$ nanoparticles on thermo-mechanical properties of the cured epoxy/TiO$_2$ nanocomposites. The dependence of the storage modulus, $G'$, loss modulus, $G''$, and loss factor, tan $\delta$, on temperature are shown in Figures 6, 7 and 8, respectively.

Figure 6. The dependence of storage modulus, $G'$, on the temperature for epoxy resin and a) TiO$_2$-OG1.0/ER, TiO$_2$-DG1.0/ER and TiO$_2$-LG1.0/ER and b) TiO$_2$-LG0.5/ER, TiO$_2$-LG1.0/ER and TiO$_2$-LG2.0/ER nanocomposites
From the results presented in Figure 6 it can be observed that all investigated samples reached rubbery plateau at temperatures higher than 100 °C. The characteristic of the rubbery plateau is that the values of the storage modulus do not depend on the temperature and therefore the $G'$ values of the examined samples determined at 120 °C are listed in Table 1. According to the result given in Figure 6 and Table 1 there is no significant difference in the values of the storage modulus between the investigated nanocomposites and pure epoxy resin in the rubbery plateau, indicating that the crosslink density of all samples is approximately the same.

The values of the glass transition temperature of pure epoxy resin and synthesized nanocomposites, determined as a maximum of the temperature dependences of $G''$ (Figure 7) and $\tan \delta$ (Figure 8), are listed in Table 1. It can be observed that nanocomposites have a higher glass transition temperature than neat epoxy resin and that the length of the hydrophobic part of gallates has no effect on the glass transition temperature of the synthesized nanocomposites, while with increasing the nanofiller content the glass transition temperature slightly increased. The obtained trend for $T_g$ values is consistent with the DSC results.

Hardness of the prepared coatings

The values of thickness and König pendulum hardness of TiO$_2$/epoxy films are shown in Table 2. These tests were carried out on cured coatings drawn on glass plates. The measured hardness values for all tested samples are approximately the same which means that the presence of the surface modified TiO$_2$ nanoparticles, as well as the change of the chain length of alkyl gallates used for surface modification of TiO$_2$ nanoparticles and their content had no significant effect on the hardness of the coatings based on epoxy resin.
Table 2. Values of the dry film thickness, König hardness and water vapor transmission rate, WVTR, of the examined samples

<table>
<thead>
<tr>
<th>Sample</th>
<th>Film thickness, μm</th>
<th>Hardness, osc</th>
<th>WVTR, g/m²/h</th>
</tr>
</thead>
<tbody>
<tr>
<td>ER</td>
<td>65</td>
<td>152</td>
<td>0.539</td>
</tr>
<tr>
<td>TiO₂-OG1.0/ER</td>
<td>65</td>
<td>149</td>
<td>0.502</td>
</tr>
<tr>
<td>TiO₂-DG1.0/ER</td>
<td>55</td>
<td>147</td>
<td>0.491</td>
</tr>
<tr>
<td>TiO₂-LG0.5/ER</td>
<td>55</td>
<td>148</td>
<td>-</td>
</tr>
<tr>
<td>TiO₂-LG1.0/ER</td>
<td>70</td>
<td>152</td>
<td>0.489</td>
</tr>
<tr>
<td>TiO₂-LG2.0/ER</td>
<td>70</td>
<td>152</td>
<td>-</td>
</tr>
</tbody>
</table>

Water vapor permeability of the synthesized TiO₂/epoxy nanocomposites

One of the basic properties of organic coatings is to protect the underlying substrate from aggressive agents, especially from water. The incorporation of inorganic particles into polymers is one of the most common approaches to improve the barrier properties of polymers. The barrier properties of nanocomposites can be improved by the adequate surface modification of nanoparticles which leads to uniform distribution of nanoparticles in the polymer matrix and a better interfacial interaction of nanoparticles and the polymer matrix. Good adhesive interface interactions give compact and a pore free microstructure, which leads to the reduction of the water vapor penetration. The strong attractive interactions at the interface increase $T_g$ and reduce free volume, contributing in this manner to better barrier properties. Also, the reduction of free volume and the improvement of barrier properties can be achieved by chemical bonding of nanoparticles to the polymer matrix [6, 34].

Figure 9. The dependence of the water transmitted through the examined films on time

Water vapor permeability of the prepared samples was investigated by the gravimetric measuring technique that determines the total water vapor transport through the epoxy films by measuring the gain of the sample mass (ASTM D1653-13). The influence of the surface modified TiO₂ nanoparticles on the permeability of the epoxy resin was tested using the permeability cup desiccant method. The amount of water, passed through the epoxy films as a function of time is shown in Figure 9. Water vapor transmission rate (WVTR) of the TiO₂/epoxy films was calculated using the slope of the linear fit of change of passed water mass vs. time. The obtained values are presented in Table 2. The water vapor transmission rate of neat epoxy resin has the highest value. Incorporation of the surface modified TiO₂ nanoparticles in epoxy resin decreased the WVTR. Also, it was observed that WVTR of TiO₂/epoxy films decreased with the increase of the alkyl chain length of gallate ligand.

Conclusions

TiO₂ nanoparticles, prepared by hydrolysis of titanium isopropoxide, were surface modified with three gallic acid esters, having different chain lengths of the hydrophobic part: octyl, decyl and lauryl gallate. The surface modified TiO₂ nanoparticles were incorporated into epoxy resin and the influence of their presence, the length of the hydrophobic part of gallates, as well as the content of lauryl gallate used for the surface modification of TiO₂ nanoparticles on glass transition temperature, rheology, barrier properties and hardness of epoxy resin was investigated.

The formation of a charge transfer complex between the TiO₂ nanoparticles and gallates was confirmed by FTIR and UV-Vis spectroscopy. Incorporation of the surface modified TiO₂ nanoparticles led to the increase of $T_g$ of epoxy resin indicating the presence of attractive interactions between the TiO₂ nanoparticles and the polymer matrix and the reduction of polymer segmental mobility at the interface between the polymer matrix and TiO₂ nanoparticles. DSC and DMA results also revealed that with increasing the nanofiller content, the glass transition temperature of the prepared nanocomposites slightly increased. The presence of the surface modified TiO₂ nanoparticles had no significant effect on the hardness of the coatings based on epoxy resin. The water vapor permeability of epoxy resin was reduced by incorporation of the surface modified TiO₂ nanoparticles, indicating that the synthesized epoxy based nanocomposites have better barrier properties than the neat epoxy resin. Furthermore, the obtained results revealed that the water vapor transmission rate of TiO₂/epoxy nanocomposites decreases with increasing the hydrophobic part chain length of gallate ligands.

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SINTEZA I KARAKTERIZACIJA NANOKOMPOZITA NA BAZI POVRŠINSKI MODIFIKOVANIH NANOČESTICA TITAN-DIOKSIDA I EPOKSIDNE SMOLE

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Ključne reči: TiO2/epoksi nanokompozit, površinska modifikacija, estri galne kiseline, propusljivost vodene pare, tvrdoća po König-u

UDK 678.743.2

Izvod


